

2001

# Order-Disorder Phase Transitions in $\text{KNO}_2$ , $\text{CsNO}_2$ , and $\text{TlNO}_2$ crystals: A Molecular Dynamics Study

Chun-Gang Duan

*University of Nebraska at Omaha, wxbdcg@gmail.com*

Wai-Ning Mei

*University of Nebraska at Omaha, physmei@unomaha.edu*

Robert W. Smith

*University of Nebraska at Omaha, robertsmith@unomaha.edu*

Jianjun Liu

*University of Nebraska-Lincoln, jliu@unlserv.unl.edu*

M. M. Ossowski

*University of Nebraska-Lincoln*

*See next page for additional authors*

Follow this and additional works at: <https://digitalcommons.unomaha.edu/physicsfacpub>

 Part of the [Chemistry Commons](#), and the [Physics Commons](#)

## Recommended Citation

Duan, Chun-Gang; Mei, Wai-Ning; Smith, Robert W.; Liu, Jianjun; Ossowski, M. M.; and Hardy, John R., "Order-Disorder Phase Transitions in  $\text{KNO}_2$ ,  $\text{CsNO}_2$ , and  $\text{TlNO}_2$  crystals: A Molecular Dynamics Study" (2001). *Physics Faculty Publications*. 15.  
<https://digitalcommons.unomaha.edu/physicsfacpub/15>

This Article is brought to you for free and open access by the Department of Physics at DigitalCommons@UNO. It has been accepted for inclusion in Physics Faculty Publications by an authorized administrator of DigitalCommons@UNO. For more information, please contact [unodigitalcommons@unomaha.edu](mailto:unodigitalcommons@unomaha.edu).



---

**Authors**

Chun-Gang Duan, Wai-Ning Mei, Robert W. Smith, Jianjun Liu, M. M. Ossowski, and John R. Hardy

# Order-disorder phase transitions in $\text{KNO}_2$ , $\text{CsNO}_2$ , and $\text{TlNO}_2$ crystals: A molecular dynamics study

Chun-gang Duan\* and W. N. Mei

*Department of Physics, University of Nebraska-Omaha, Nebraska 68182-0266*

R. W. Smith

*Department of Chemistry, University of Nebraska-Omaha, Nebraska 68182-0109*

Jianjun Liu, M. M. Ossowski, and J. R. Hardy

*Department of Physics and Center for Electro-optics, University of Nebraska-Lincoln, Nebraska 68588-0111*

(Received 5 January 2000; revised manuscript received 14 November 2000; published 16 March 2001)

The order-disorder phase transitions of  $\text{KNO}_2$ ,  $\text{CsNO}_2$ , and  $\text{TlNO}_2$  have been studied using parameter-free molecular dynamics simulation. It is found that the phase transitions in nitrites investigated are driven by the rotations of the  $\text{NO}_2^-$  ions about different axes together with displacements of cations and anions. We successfully reproduce the high-temperature phases of these nitrites, i.e., the NaCl-like structure for  $\text{KNO}_2$  and CsCl-like structure for  $\text{Cs(Tl)NO}_2$ . Based on the investigation of the radial distribution function of the cations and anions, we explain why  $\text{KNO}_2$  and  $\text{Cs(Tl)NO}_2$  form quite different low-temperature phases.

DOI: 10.1103/PhysRevB.63.144105

PACS number(s): 64.60.Cn, 61.43.Bn, 64.70.Pf

## I. INTRODUCTION

The high-temperature phase of potassium nitrite and the room-temperature phases of cesium and thallium nitrite are known as plastic crystals because they are mechanically soft. This special property is believed to be associated with the disordered orientations of the  $\text{NO}_2^-$  anions in the lattice. Experimental studies on these nitrites have been carried out extensively by a variety of techniques.<sup>1-7</sup>

$\text{KNO}_2$  in phase I ( $T > 315$  K) has NaCl structure, with spherically disordered orientations of the  $\text{NO}_2^-$  ions. At lower temperature ( $264$  K  $< T < 315$  K), in phase II, the structure is rhombohedral (space group  $R\bar{3}m$ ) with orientations of  $\text{NO}_2^-$  ions disordered about the rhombohedral  $\bar{3}$  axis. In phase III ( $T < 264$  K), it has a monoclinic structure (space group  $P2_1/c$ ) with ordered  $\text{NO}_2^-$  ions.<sup>5</sup>

$\text{CsNO}_2$  and  $\text{TlNO}_2$  both have CsCl-type structures in their room-temperature phase, phase I, in which the  $\text{NO}_2^-$  ions are likely to be in free rotation as in phase I of  $\text{KNO}_2$ . They undergo phase transitions at 209.2 and 282.4 K, respectively, to phase II, which has a trigonal structure with space group  $P3_121$ .<sup>7</sup>

Recently, neutron powder diffraction studies on,  $\text{KNO}_2$  by Onoda-Yamamuro *et al.*,<sup>5</sup> and on  $\text{CsNO}_2$  and  $\text{TlNO}_2$  by Feyerherm, Collins, and Johari,<sup>7</sup> have thrown new light on the low-temperature phases of these crystals. Specifically, it was found that, at very low temperature, the  $\text{NO}_2^-$  ions in these three nitrites are highly ordered, but there exists a small but distinct development of the orientational disorder of the  $\text{NO}_2^-$  ions with increasing temperature. This can be associated, at least for  $\text{CsNO}_2$  and  $\text{TlNO}_2$ , with glasslike behaviors in specific-heat and dielectric relaxation measurements.<sup>3,4,6</sup>

The nature of the phase-transition mechanism is still an open question. It is very interesting to carry out a theoretical study to clarify the mechanism governing the order-disorder phase transitions and the behavior of the  $\text{NO}_2^-$  ions in the

low-temperature phase of these nitrites. Molecular dynamics (MD) simulation provides a direct way to answer such questions. One can obtain a great deal of information about the thermal and microscopic behaviors of the ions in the crystal by MD simulation. Specifically, in the past, we successfully simulated the structural phase transition in  $\text{NaNO}_2$ .<sup>8</sup> In this paper we report the results of simulations on  $\text{KNO}_2$ ,  $\text{CsNO}_2$ , and  $\text{TlNO}_2$  using the same technique. Here we focus on the behavior of nitrite ions before and after phase transitions.

## II. POTENTIAL-ENERGY SURFACE AND STATIC RELAXATION

Our MD simulation technique was developed by Lu and Hardy to study phase transitions in ionic molecular crystals.<sup>9</sup> This method originates from the Gordon-Kim model for ionic crystal (e.g., alkali halides).<sup>10</sup> It was extended to deal with ionic molecular crystals like  $\text{K}_2\text{SeO}_4$ , in which strong intermolecular covalency exists. The main idea is that in ionic molecular crystal the molecular ion ( $\text{SeO}_4^{2-}$  in  $\text{K}_2\text{SeO}_4$ ) is treated as a single entity, and intramolecular and intermolecular interactions are considered separately. First we perform *ab initio* quantum-chemistry calculations for the whole molecular ion, to obtain the optimized structure and the electron charge densities of each individual atom in the molecular ion. Then the intermolecular interactions are calculated from the resultant charge densities together with those of the cation by using the Gordon-Kim electron-gas model, while the intramolecular interactions are described by an *ab initio* harmonic expansion of the energy of the molecular ion. This method provides a parameter-free description for the crystal potential-energy surfaces which allow structural relaxation, MD simulation, and lattice dynamics calculations.

In this work, we used the GAUSSIAN 94 program package<sup>11</sup> to perform *ab initio* quantum-chemistry structural optimization for the nitrite ion. Consistent with earlier work on

TABLE I. Experimental and theoretical structural parameters for the  $\text{NO}_2^-$ . Theoretical values are obtained from using GAUSSIAN 94 package.

	N-O distance (Å)	O-N-O angle
KNO <sub>2</sub>	1.250 (N-O1)	
	1.256 (N-O2)	116.9°
CsNO <sub>2</sub>	1.23	114.8°
TlNO <sub>2</sub>	1.25	113.8°
Theory	1.23	116.6°

NaNO<sub>2</sub>, we used the double-zeta basis for the nitrogen and oxygen atoms to obtain the optimized structure of the nitrite ion. The parameters of the optimized structure, together with the experimental values, are listed in Table I. The Mulliken population analysis of the *ab initio* charge density gives charges of 0.16256e and -0.58128e for N and O atoms, respectively. These charges, together with cationic charges of +1e, were used to calculate the long-range Coulomb interactions. In the calculation of the Gordon-Kim pair potentials, the free ion charge density for K<sup>+</sup> was taken from Ref. 12, and those for Cs<sup>+</sup> and Tl<sup>+</sup> from Ref. 13.

We then used the potentials obtained above to perform static relaxations with crystal structure symmetry constraints. The relaxation started from experimental structures, and periodic boundary conditions and a Newton-Raphson algorithm were used. These relaxations provide crystal structures in which the forces on the ions and the stresses are zero. The parameters of relaxed structure along with the experimental values are listed in Table II. The relaxed structure of KNO<sub>2</sub> with space group  $P2_1/c$  constraints is not in good agreement with experiments. However, if the monoclinic angle  $\beta$  was kept constant at the experimental value during relaxation, the final structure, which is listed in Table II under the column

TABLE II. Experimental and theoretical structural parameters for the  $P2_1c$  phase of KNO<sub>2</sub>. (Lattices constants are given in Å.)

Parameters	Experiment	Theory	Theory (fix $\beta$ )
<i>a</i>	4.4010	4.0418	4.2184
<i>b</i>	9.5967	7.7796	8.1846
<i>c</i>	6.9817	8.1876	6.9715
$\beta$	108.274°	119.580°	108.274°
<i>x/a</i> of N	-0.0732	-0.1130	0.0018
<i>y/b</i> of N	0.4107	0.4462	0.3869
<i>z/c</i> of N	0.1839	0.1370	0.1399
<i>x/a</i> of K	0.4105	0.4278	0.3783
<i>y/b</i> of K	0.1407	0.1619	0.1112
<i>z/c</i> of K	0.2101	0.1778	0.2201
<i>x/a</i> of O(1)	0.2034	0.0575	0.1497
<i>y/b</i> of O(1)	0.4127	0.4204	0.4115
<i>z/c</i> of O(1)	0.3067	0.3075	0.3200
<i>x/a</i> of O(2)	-0.1615	-0.2120	-0.2822
<i>y/b</i> of O(2)	0.2999	0.3168	0.3316
<i>z/c</i> of O(2)	0.0865	0.0380	0.1009

“Theory (fix  $\beta$ ),” is close to the experiment except for a contraction of the lattice parameters of  $\sim 5\%$ . The agreements between theoretical and experimental structures of CsNO<sub>2</sub> and TlNO<sub>2</sub> are much better, with the lattice parameters 8% shorter than the experimental values. This is a general feature of simulations using Gordon-Kim potentials.<sup>9</sup>

The discrepancies between the theoretical and experimental values of the structural parameters for KNO<sub>2</sub>, CsNO<sub>2</sub>, and TlNO<sub>2</sub> are larger than those for NaNO<sub>2</sub>.<sup>8</sup> We attribute this to the large number of variables and thus more complex potential surfaces of the nitrites investigated here. This allows more possibilities for locally metastable states than is the case for the NaNO<sub>2</sub>  $Im2m$  structure. In particular, for KNO<sub>2</sub>, the major discrepancies are probably due to relatively large rotations of the  $\text{NO}_2^-$ , inducing compensatory changes in  $\beta$  to produce only a small change in the overall energy. However, since our main goal focuses on the nature of the order-disorder phase transitions in these nitrites, we consider the agreements between theory and experiment adequate to proceed with the MD simulation study.

### III. MD SIMULATIONS OF THE PHASE TRANSITIONS

We performed constant (zero) pressure MD calculations<sup>14</sup> to simulate the temperature-dependent behaviors of these nitrites. During the simulation, both the atomic positions and the lattice vectors were allowed to change. Periodic boundary conditions were imposed to provide an infinite-size lattice. Supercells were chosen large enough to permit all possible phase transitions to occur. The systems were heated and cooled both in 20-K step. At each stage the averages of the atomic positions, internal energy, etc., were taken over a period of 30 ps using a 0.003-ps MD time step.

#### A. KNO<sub>2</sub>

We used a supercell with 432 ions formed by tripling the  $P2_1/c$  cell in all three directions. The simulation was started from the relaxed structure obtained by fixing the monoclinic  $\beta$  angle to experimental value and the sample was heated up to 600 K. Figure 1 presents the specific heat  $C_p$  as a function of temperature obtained from our simulation. As can be seen from the figure, there is one sharp peak at nearly 380 K, which indicates the occurrence of one phase transition. This peak is about 60 K higher than the experimental phase II-I transition temperature, 315 K. We also found a small but noticeable shoulder at around 340 K, which we attribute to be an indication of another phase transition. The temperature difference (40 K) of the two theoretical phase transitions is close to the experimentally observed value of 50 K.

To study the mechanism of these phase transitions, the average crystal structures of KNO<sub>2</sub> at several different temperatures are monitored and depicted in Fig. 2. The ellipsoids in these figures represent the rms deviations of the atoms from their average positions, and thus indicate the thermal motions of these atoms. For clarity, only one certain layer of atoms is shown. At a low temperature (about 120 K), we observed that some nitrite ions flip around an axis parallel to the O-O bond in the nitrite. As the temperature increases,

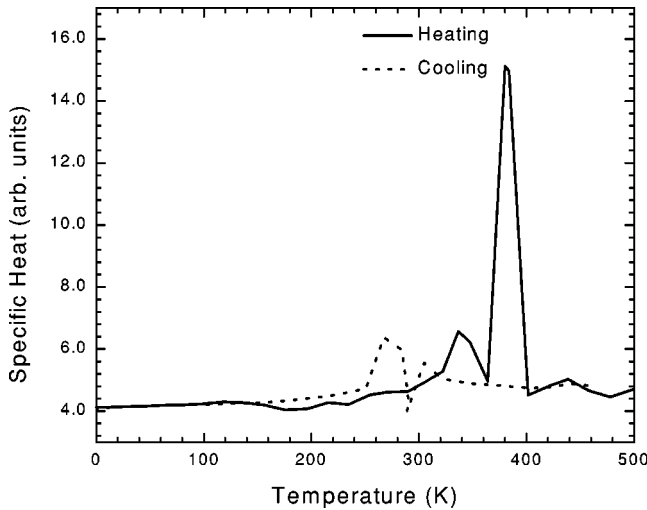


FIG. 1. Temperature variation of the specific heat  $C_p$  of  $\text{KNO}_2$  obtained from molecular dynamics simulation heating and cooling processes.

both phase transitions occur when the nitrite ions begin to rotate about a new axis. The orientations of the nitrite ions are completely disordered after 380 K, and the system becomes a sodium chloride structure.

Starting from a heated disordered structure at about 460 K, we also performed a MD cooling process for  $\text{KNO}_2$ . On the specific-heat curve (indicated as ‘‘Cooling’’ in Fig. 1) we also find two peaks in the temperature range 260–360 K. Again, average structure studies reveal that they are associated with the freezing of different kinds of rotational motions.

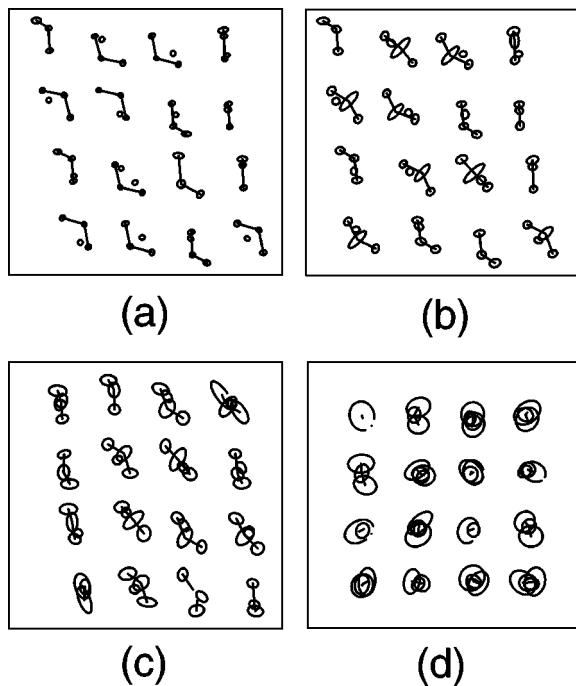


FIG. 2. Atomic positions of  $\text{KNO}_2$  viewed from the  $\mathbf{a}$  direction at (a)  $T=118$  K, (b)  $T=136$  K, (c)  $T=347$  K, and (d)  $T=384$  K.

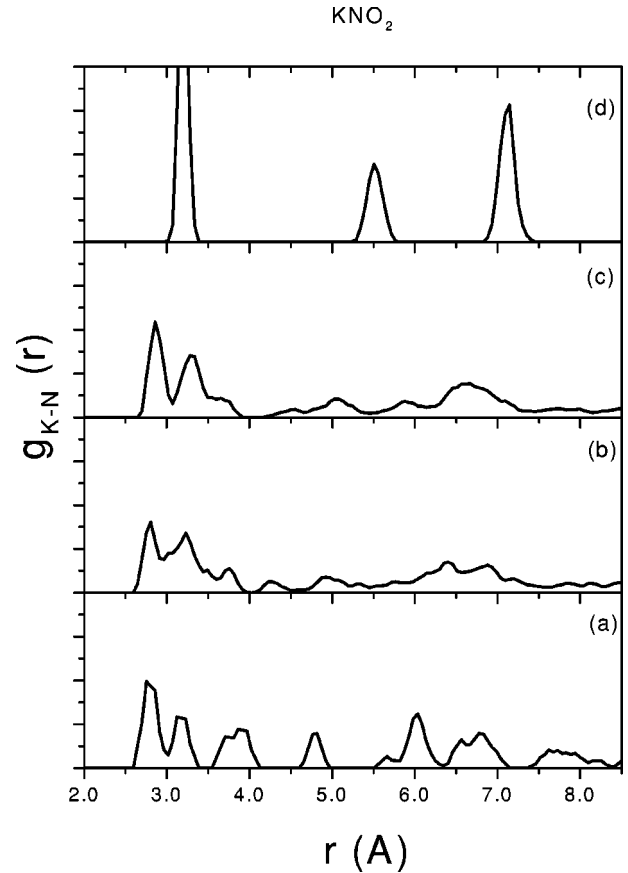


FIG. 3. K-N radial distribution function at (a) 20 K, (b) 321 K, (c) 347 K, and (d) 500 K.

Now it is interesting to make clear the structure of the intermediate phase, and the structural relationship between each phase. To do this, we plot the radial distribution function  $g(r)$  of the K and N atoms at several different temperatures in Fig. 3. The average structures at these temperatures are obtained from the heating run. It can be seen from these figures that there is no significant structure change from 321 to 347 K. Thus, in our simulation, this phase transition is mainly caused by the occurrence of a new mode of rotation of nitrite ions.

The three sharp peaks appeared in panel (d) of Fig. 3 indicate clearly that the high-temperature phase of  $\text{KNO}_2$  is a NaCl-like phase. The relative ratios of these peak positions agree very well with those of a NaCl structure. A noticeable structure change also can be found from low temperature to the first phase transition temperature. However, the integration over the region from 2.5 to 4.1 Å remains at 6, which is the correct number of the nearest-neighboring N atoms of K atom in a NaCl-like structure. This indicates that, when cooling from a high-temperature cubic structure, the K atom’s neighboring N atoms remain almost unchanged.

We find that the monoclinic phase III is a distorted structure of rhombohedral phase II. Both K and N atom form a distorted face-centered lattice. As the temperature increases, the nitrite ions become orientationally disordered, and in at the same time the K and N atoms make some translations to form an ideal rhombohedral face-centered lattice at the first

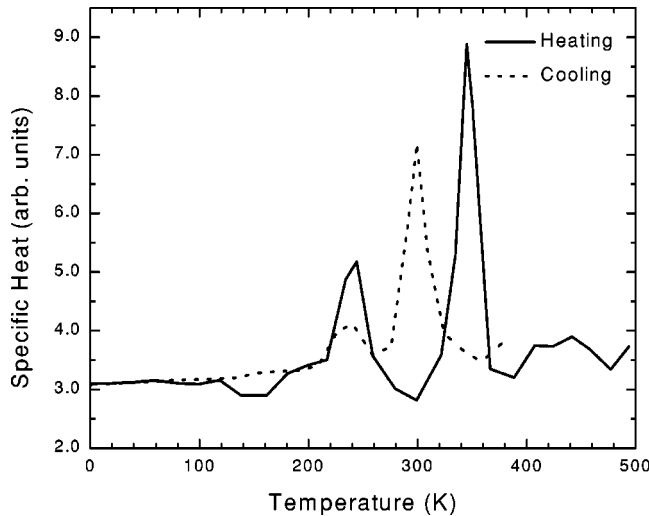


FIG. 4. Temperature variation of the specific heat  $C_p$  of  $\text{CsNO}_2$  obtained from molecular dynamics simulation heating and cooling processes.

phase transition. However, the translations of the  $\text{K}^+$  and  $\text{NO}_2^-$  ions are somewhat delicate. Thus an exact reproduction of the phase III-II transition seems go beyond the time scales accessible to our MD simulations. On the other hand, the rhombohedral lattice of phase II can be described as a distorted NaCl structure. The distortion is due to dilation along one of the four cubic diagonals of the NaCl structure. When the temperature becomes higher, the rotation of the nitrite ions became almost free. Thus, at the second phase transition, the rhombohedral structure contracts along its threefold axis to become NaCl structure.

### B. $\text{CsNO}_2$ and $\text{TlNO}_2$

Supercells with 324 ions of  $\text{CsNO}_2$  and  $\text{TlNO}_2$  were both constructed by tripling the  $P3_121$  cell in all three directions. The samples were heated from a relaxed structure up to about 500 K.

On the specific-heat curve obtained from the heating process for  $\text{CsNO}_2$  (Fig. 4), we find two distinct peaks at 240 and 340 K, which does not agree with experiments. However, based on the average structure analysis, we note that in our simulation, the phase transition from phase II to phase I is conducted in two steps. These correspond to two different kinds of rotations of  $\text{NO}_2^-$  ions which begin from nearly 220 and 330 K, respectively. Apparently these two steps occur at the same time in the experiment. By studying the radial distribution functions of the Cs and N atoms at several different temperatures (Fig. 5), we found that the first theoretical phase transition does not lead to any significant structural change. It is only caused by an of additional rotational mode taking place, i.e., the increase of the degree of disorder of nitrite ions. As can be seen from the figure, the features of these curves, in sharp contrast to those of  $\text{KNO}_2$ , almost remain unchanged throughout the entire MD simulation. The integration over the region from 3.0 to 4.5 Å remains nearly at 8, which is the correct number of the N-atoms that are nearest neighbors to the Cs atom in the CsCl-like structure.

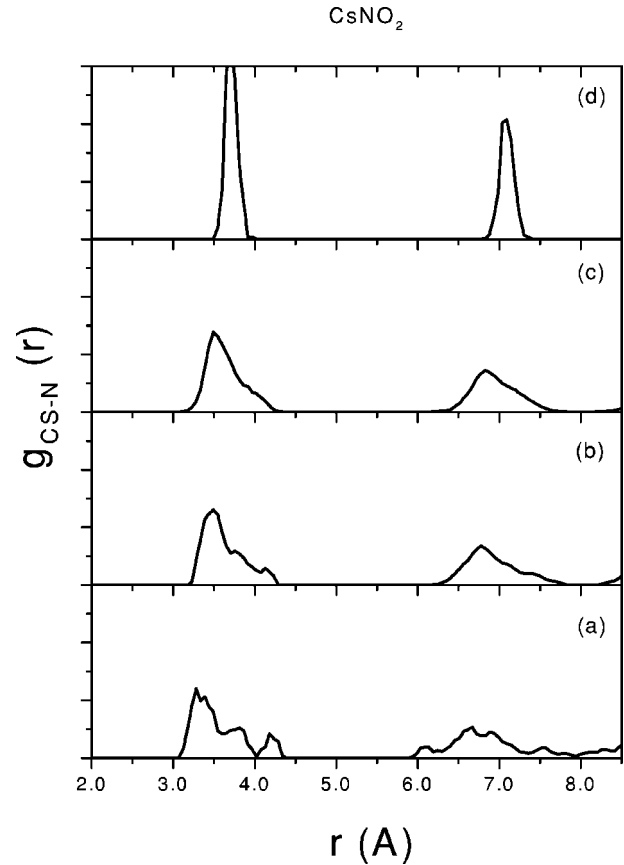


FIG. 5. Cs-N radial distribution function at (a) 20 K, (b) 234 K, (c) 259 K, and (d) 500 K.

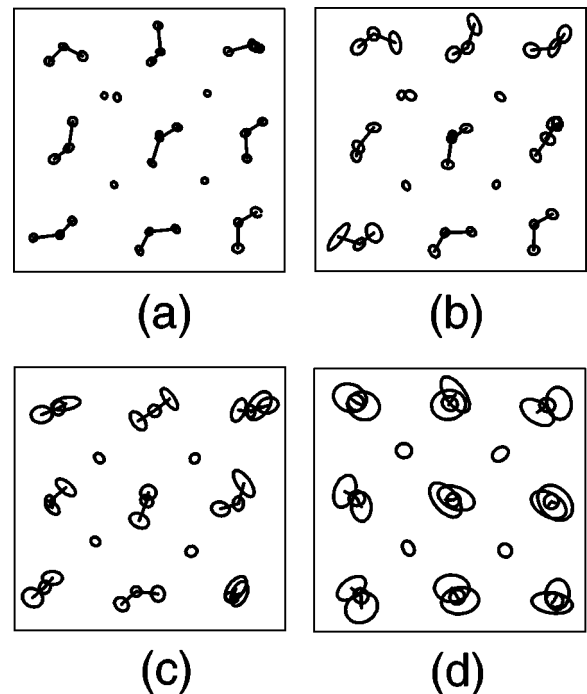


FIG. 6. Atomic positions of  $\text{CsNO}_2$  viewed from the  $\mathbf{b}$  direction at (a)  $T=118$  K, (b)  $T=138$  K, (c)  $T=244$  K, and (d)  $T=350$  K.



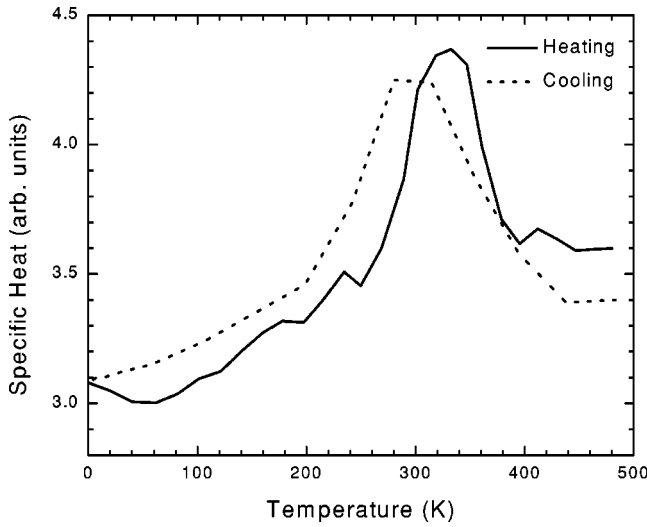


FIG. 7. Temperature variation of the specific heat  $C_p$  of  $\text{TiNO}_2$  obtained from molecular dynamics simulation heating and cooling processes.

One can see from Fig. 6 that below 120 K the system remains a  $P3_121$  structure. With an increase in the temperature, the  $\text{NO}_2^-$  ions begin to flip around an axis parallel to the O-O bond in the nitrite. At about 240 K, another kind of rotation of  $\text{NO}_2^-$  ions occurs indicating the onset of a phase transition. As the temperature reaches 340 K, the  $\text{NO}_2^-$  ions rotate freely and the orientations of  $\text{NO}_2$  ions become completely disordered. Then the structure transforms to CsCl-type structure with some displacement of Cs atoms.

We also cooled the sample from disordered structures at about 380 K down to very low temperature. In that specific heat curve (see Fig. 4) we find two peaks at 230 and 300 K, respectively. Both of these are lower than their counterparts in the heating run, but the average structure studies still show that they correspond to a freezing of the rotation of the nitrite anions.

For  $\text{TiNO}_2$ , we note that the system was stable at an experimental melting temperature 465 K. We attribute this to the intrinsic limitation that our simulation was performed on a sample without free surfaces. As distinct from the case of  $\text{CsNO}_2$ , no two-peak structure is found in the specific-heat curves (Fig. 7). In fact, the phase transition is rather “sluggish,” which begins at  $\sim 250$  K and ends at  $\sim 360$  K. At low temperature, the behaviors of the nitrite ions are almost the same as those of  $\text{CsNO}_2$ . Therefore, we observed the same kind of flipping motion of nitrite ions. Analogous to our understanding of the phase transition of  $\text{CsNO}_2$ , the broad phase transition of  $\text{TiNO}_2$  can be interpreted as a continuous unlocking (or freezing when cooling) of the rotational degrees of freedom. As was the case for  $\text{CsNO}_2$ , the high-temperature phase is a CsCl-type cubic structure (see Table III).

The evolution of the average structures of  $\text{CsNO}_2$  and  $\text{TiNO}_2$  with temperature confirms the relationship between phases I and II proposed by Feyerherm *et al.*<sup>7</sup> The Cs (Ti) and N atoms form a distorted CsCl lattice in phase II. Starting from the trigonal structure of phase II, the Cs (Ti) and N

TABLE III. Experimental and theoretical structural parameters for the  $P3_121$  phases of  $\text{CsNO}_2$  and  $\text{TiNO}_2$ . (Lattices constants are given in Å.)

Parameters	Experiment	Theory
$\text{CsNO}_2$		
$a (=b)$	5.8670	5.4348
$c$	7.8304	7.8942
$x/a$ of N	0.3122	0.1964
$x/a$ of Cs	0.3053	0.4000
$x/a$ of O(1)	0.3755	0.3567
$y/b$ of O(1)	0.4768	0.2754
$z/c$ of O(1)	0.6147	0.6229
$\text{TiNO}_2$		
$a (=b)$	5.5490	5.4022
$c$	7.3233	7.9702
$x/a$ of N	0.3098	0.2349
$x/a$ of Ti	0.2973	0.3655
$x/a$ of O(1)	0.3715	0.3570
$y/b$ of O(1)	0.4940	0.3530
$z/c$ of O(1)	0.6182	0.6309

atoms first are slightly displaced, to form a rhombohedral structure. Then the rhombohedral structure contracts along its threefold axis to become a CsCl structure. These two steps take place together at the phase transition.

#### IV. DISCUSSIONS

Monovalent nitrites  $M\text{NO}_2$  ( $M=\text{Na}, \text{K}, \text{Cs}, \text{Ti}$ ) have all been found to have ordered structures in their low-temperature phases. These crystals, except  $\text{NaNO}_2$ , all crystallize into plastic phases at room temperature because of the orientational disorder of the nitrite ions.

As presented in Sec. III, all the phase transitions in the nitrites investigated are related to the rotation of the nitrite ions, about different axes. One can argue that three possible rotations of the nitrite ion might produce three sequential phase transitions. The first kinds of rotation (a) is about an axis in the plane of nitrite ions and parallel to the O-O axis. The other two kinds of rotations are about the twofold axis in the  $\text{NO}_2$  plane (b) and about an axis perpendicular to the nitrite plane (c).

Using low-temperature experimental structures, we calculated the potential energies as a function of the angle of rotation of one of the nitrites in the unit cell in the above three modes. The rotational mode (a) has a energy barrier 5–10 times smaller than those of the other two kinds of rotations. It is expected to be much easier to excite than the other two modes. Thus we observe this kind of motion at relatively low temperatures. Mode (a) only slightly changes the total energy of the whole system, but it does produce an excess entropy and thus affects the thermal properties, e.g., the specific heat of the crystal. This kind of rotation may be the reason for the glass-transition-like behavior of these nitrites in the low-temperature phases. Feyerherm and co-workers<sup>7,15</sup> found that at low temperature  $\text{CsNO}_2$  and

TiNO<sub>2</sub> only have a small-angle orientational disorder and small-scale reorientational motions of the NO<sub>2</sub> anions, and suggested that the NO<sub>2</sub> ions liberate about an axis other than the  $C_{2v}$  axis. The logical candidate is mode (a), the libration of NO<sub>2</sub> ions about an axis parallel to the O-O line.

The other two kinds of rotations, namely, modes (b) and (c), have much higher energy barriers than that of mode (a), and require more energy to excite them. Thus they occur at higher temperature. These two kinds of rotation are so drastic that they can result in a peak in the  $C_p$  curve.

The order-disorder phase transition in a molecular ion crystal often involves a displacements of cations. At high temperature, the molecular anions go into nearly free rotations and become effectively spherical. In a CsCl structure the cations have to be sufficiently large for the first neighbor. This simple cubic is inherently unstable and will shear into a fcc structure. This process will only be prevented if the cations' radii exceed a threshold value such that the first-neighbor anion-cation repulsion is sufficient to keep the anions out of effective contact. Thus molecular crystals with smaller cations usually have a NaCl structure, e.g., RbNO<sub>3</sub> and RbN<sub>3</sub>, while those with larger cations form CsCl lattices, e.g., CsNO<sub>3</sub>, TiNO<sub>3</sub>, and CsN<sub>3</sub>. Analogously, KNO<sub>2</sub> has NaCl structures and CsNO<sub>2</sub> and TiNO<sub>2</sub> have CsCl structure at high temperature because of their different cation sizes. As the temperature decreases, the small K atom cases the coordination number to remain small, whereas large Cs and Tl atoms still need a larger coordination number. This

can explain why KNO<sub>2</sub> has a different low-temperature structure from that of CsNO<sub>2</sub> and TiNO<sub>2</sub>.

## V. CONCLUSIONS

We have studied, for the first time at the microscopic level to our knowledge, the order-disorder phase transitions of KNO<sub>2</sub>, CsNO<sub>2</sub>, and TiNO<sub>2</sub> by using a parameter-free MD simulation method. By studying the pair relationship of the cations and anions at different temperatures of these nitrites, we explained why these nitrites, though having the same anionic groups, have different low-temperature phases. We found that cations play a very important role in forming the structure of these crystals, i.e., the phase diagram can be very different because of the different sizes of the cations. In our simulations all the phase transitions occurring in these nitrite are driven mainly by different rotational motions of NO<sub>2</sub><sup>-</sup> anions. We also found that the glass-transition-like behavior of these nitrites is possibly caused by liberations of the nitrite ions around an axis parallel to the O-O bond.

## ACKNOWLEDGMENTS

We thank Professor M. F. Collins for sending us his recent publication. This work was supported by the U.S. Army Research Office under (a) Grants No. DAAG 55-97-1-0106 and (b) Grants No. DAAG 55-98-1-0273. The computer facility was supported by Nebraska-EPSCoR-NSF Grant No. EPS-9720643.

\*Email: dcg@cosmos.unomaha.edu.

<sup>1</sup>J. K. Solbakk and K. O. Strømme, *Acta Chem. Scand.* **23**, 300 (1969); K. O. Strømme, *Z. Anorg. Allg. Chem.* **403**, 176 (1974).

<sup>2</sup>Carl W. F. T. Pistorius and P. W. Richter, *Z. Anorg. Allg. Chem.* **389**, 315 (1972).

<sup>3</sup>S. C. Mraw, R. J. Boak, and L. A. K. Staveley, *J. Chem. Thermodyn.* **10**, 359 (1978).

<sup>4</sup>K. Moriya, T. Matsuo, and H. Suga, *J. Phys. Chem. Solids* **44**, 1103 (1983).

<sup>5</sup>N. Onoda-Yamamuro *et al.*, *J. Phys.: Condens. Matter* **10**, 3341 (1998).

<sup>6</sup>H. Honda *et al.*, *Ber. Bunsenges. Phys. Chem.* **102**, 148 (1998).

<sup>7</sup>R. Feyerherm, M. F. Collins, and G. P. Johari, *Phys. Rev. B* **57**, 11 125 (1998).

<sup>8</sup>H. M. Lu, R. Qi, and J. R. Hardy, *Solid State Commun.* **87**, 1151 (1993).

<sup>9</sup>H. M. Lu and J. R. Hardy, *Phys. Rev. Lett.* **64**, 661 (1990); *Phys. Rev. B* **42**, 8339 (1990).

<sup>10</sup>R. G. Gordon and Y. S. Kim, *J. Chem. Phys.* **56**, 3122 (1972).

<sup>11</sup>M. J. Frisch *et al.*, *Gaussian 94 User's Reference* (Gaussian, Inc., 1994).

<sup>12</sup>E. Clementi and C. Roetti, *At. Data Nucl. Data Tables* **14**, 177 (1974).

<sup>13</sup>A. D. McLean and R. S. McLean, *At. Data Nucl. Data Tables* **26**, 197 (1981).

<sup>14</sup>M. Parrinello and A. Rahman, *Phys. Rev. Lett.* **45**, 1196 (1980).

<sup>15</sup>G. P. Johari, M. F. Collins, and R. Feyerherm, *Phys. Rev. B* **61**, 5827 (2000).